

A STRUCTURAL STUDY OF THE LITHIOPHILITE–SICKLERITE SERIES

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ABSTRACT

A sample from the Koptokay No. 3 granitic pegmatite, Altai Mountains, northwestern China, which shows a transition from lithiophilite [$\text{LiMn}^{2+}\text{PO}_4$] to sicklerite [$\text{Li}_{1-x}\text{Mn}^{2+}\text{PO}_4$], was investigated by single-crystal X-ray diffraction, electron-microprobe analysis, and secondary-ion mass spectrometry (SIMS) techniques. Under the polarizing microscope, the sample shows colorless lithiophilite and deep-orange sicklerite, as well as several intermediate phases. The chemical compositions change from $\text{Li}_{0.96}(\text{Mn}^{2+}_{0.81}\text{Fe}^{2+}_{0.09}\text{Fe}^{3+}_{0.08})\text{PO}_4$ to $\text{Li}_{0.69}(\text{Mn}^{2+}_{0.62}\text{Mn}^{3+}_{0.19}\text{Fe}^{3+}_{0.16})\text{PO}_4$, and show a progressive decrease of the lithium content from lithiophilite to sicklerite. Five crystals were extracted from the thin section, and their crystal structures were refined in space group *Pbnm*, with unit-cell parameters from a 4.736(1), b 10.432(2), c 6.088(1) Å (lithiophilite), to a 4.765(1), b 10.338(1), c 6.060(1) Å (sicklerite). The olivine-type structures of these phosphates are identical to that of triphylite, and are characterized by two chains of edge-sharing octahedra parallel to the *c* axis. The first chain consists of *M1* octahedra containing Li atoms and vacancies, and the second chain consists of *M2* octahedra occupied by Fe and Mn. The significant Mn^{3+} content in sicklerite necessitates a more careful interpretation of the electron-microprobe data, and demonstrates the different mechanisms of oxidation affecting the lithiophilite–sicklerite and triphylite–ferrisicklerite series, respectively.

Keywords: olivine-type phosphates, lithiophilite, sicklerite, crystal chemistry.

SOMMAIRE

Un échantillon de la pegmatite Koptokay No. 3, Altaï, République populaire de Chine, a été étudié par diffraction des rayons X sur monocristaux, et a fait l'objet d'analyses chimiques à la microsonde électronique et à la sonde ionique (SIMS). Cet échantillon montre une transition progressive entre la lithiophilite incolore [$\text{LiMn}^{2+}\text{PO}_4$] et la sicklérite orange [$\text{Li}_{1-x}\text{Mn}^{2+}\text{PO}_4$], accompagnée de plusieurs phases intermédiaires. Les analyses chimiques indiquent des compositions comprises entre $\text{Li}_{0.96}(\text{Mn}^{2+}_{0.81}\text{Fe}^{2+}_{0.09}\text{Fe}^{3+}_{0.08})\text{PO}_4$ et $\text{Li}_{0.69}(\text{Mn}^{2+}_{0.62}\text{Mn}^{3+}_{0.19}\text{Fe}^{3+}_{0.16})\text{PO}_4$, et montrent une diminution progressive du contenu en lithium, lorsque l'on passe de la lithiophilite à la sicklérite. Cinq monocristaux ont été extraits de la lame mince, et leurs structures cristallines ont été affinées dans le groupe spatial *Pbnm*, avec des paramètres de maille compris entre a 4,736(1), b 10,432(2), c 6,088(1) Å (lithiophilite) et a 4,765(1), b 10,338(1), c 6,060(1) Å (sicklérite). La structure de type olivine de ces phosphates est identique à celle de la triphylite, et est constituée de deux chaînes octaédriques parallèles à l'axe cristallographique *c*. La première chaîne contient les octaèdres *M1* occupés par le lithium et des lacunes, alors que la seconde chaîne contient les octaèdres *M2* occupés par le fer et le manganèse. La présence de quantités non négligeables de Mn^{3+} dans la sicklérite démontre la nécessité d'interpréter les analyses chimiques à la microsonde électronique avec précaution, et indique également que les processus d'oxydation affectant les solutions solides lithiophilite–sicklérite et triphylite–ferrisicklérite sont génétiquement distincts.

Mots-clés: phosphates à structure olivine, lithiophilite, sicklerite, cristalochimie.

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INTRODUCTION

Iron–manganese phosphates are common accessory minerals in granitic pegmatites, in metamorphic rocks, and in meteorites. In rare-element pegmatites, primary phosphates of the triphylite–lithiophilite series $[\text{Li}(\text{Fe}^{2+}, \text{Mn}^{2+})(\text{PO}_4) - \text{Li}(\text{Mn}^{2+}, \text{Fe}^{2+})(\text{PO}_4)]$ form masses that can reach several meters in diameter, enclosed in silicates. During the oxidation processes affecting the pegmatites, these olivine-type phosphates transform to ferrisicklerite – sicklerite $[\text{Li}_{1-x}(\text{Fe}^{3+}, \text{Mn}^{2+})(\text{PO}_4) - \text{Li}_{1-x}(\text{Mn}^{2+}, \text{Fe}^{3+})(\text{PO}_4)]$ and heterosite – purpurite $[(\text{Fe}^{3+}, \text{Mn}^{3+})(\text{PO}_4) - (\text{Mn}^{3+}, \text{Fe}^{3+})(\text{PO}_4)]$ by the substitution mechanism $\text{Li}^+ + \text{Fe}^{2+} \rightarrow \square + \text{Fe}^{3+}$. This oxidation sequence was first observed by Quensel (1937) and confirmed by Mason (1941).

Olivine-type phosphates crystallize in space group *Pbnm* (triphylite: a 4.690, b 10.286, c 5.987 Å), and their crystal structures have been investigated with synthetic samples (Geller & Durand 1960, Yakubovich *et al.* 1977) and minerals (Finger & Rapp 1969, Eventoff *et al.* 1972, Moore 1972, Alberti 1976, Losey *et al.* 2004, Fehr *et al.* 2007). Since Padhi *et al.* (1997) reported the reversible electrochemical extraction of lithium from LiFePO_4 , olivine-type phosphates LiMPO_4 ($M = \text{Fe}, \text{Mn}, \text{Co}, \text{Ni}$) have received major attention as candidates for lithium batteries (*e.g.*, Andersson *et al.* 2000, Okada *et al.* 2001, Ravet *et al.* 2003, Proisini *et al.* 2002, Song *et al.* 2002, Takahashi *et al.* 2002, Yang *et al.* 2003, Bramnik *et al.* 2004, Deniard *et al.* 2004, Fehr *et al.* 2007). Recently, Kang & Ceder (2009) have shown that LiFePO_4 -based batteries can achieve ultrarapid charging and discharging, in 10–20 s, thus reaching the performance of supercapacitors. These exceptional features place LiFePO_4 among the best candidates for producing batteries for many applications, such as electric bicycles, electric boats, electric cars, or for the storage of green energy. In order to elucidate the structural modifications induced by the oxidation processes affecting lithiophilite, we decided to investigate a natural sample from the Altai Mountains, China, in which a progressive transition from lithiophilite to sicklerite is observed.

SAMPLE DESCRIPTION AND GEOLOGICAL SETTING

The sample investigated herein (#K9–5–6) was collected by Dr. François Fontan in the Koktokay No. 3 granitic pegmatite, Altai Mountains, northwestern China. It consists of a hand specimen of $7 \times 6 \times 4$ cm in size, containing quartz, spodumene, lithiophilite and sicklerite. A section in the sample shows a large (3×1 cm) crystal of yellowish lithiophilite surrounded by a brownish rim *ca.* 5 mm thick, consisting of sicklerite. In thin section, the lithiophilite is colorless, and the sicklerite is deep orange. Moreover, several grains located between lithiophilite and sicklerite exhibit intermediate colors, ranging from pale yellow to yellow or orange.

The contacts between these grains are generally sharp, and correspond to the cleavage planes or to fractures, thus giving the grains unusual angular shapes (Fig. 1a).

Roughly 100,000 pegmatite dykes occur in the Chinese Altai pegmatite field; the Koktokay No. 3 pegmatite is one of the most strongly zoned. This pegmatite, which shows a NW–SE elongation, intrudes a metagabbro located in gneissic biotite schists (Zhang *et al.* 2004), and was formed during the Yanshanian period (177.9–148.0 million years; Chen *et al.* 2000). On the basis of their mineral contents, nine zones were described in the pegmatite, and the sample investigated in this study was collected in zone V, characterized by the presence of bladed albite and spodumene (Zhu *et al.* 2000). Lithiophilite and sicklerite from this zone

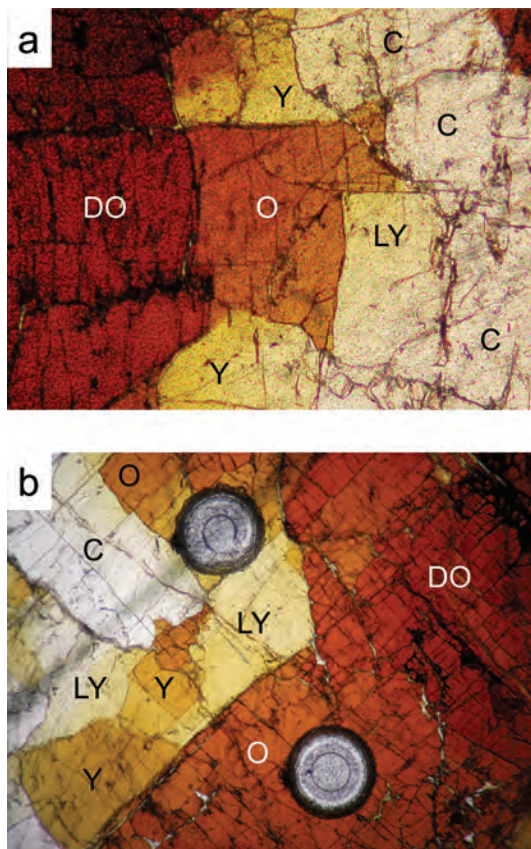


FIG. 1. a. Micrograph showing the transition from lithiophilite (colorless) to sicklerite (deep orange), with several grains exhibiting intermediate colors. b. Thin section after drilling in the orange and yellow zones. Notation: C: colorless lithiophilite, LY: light yellow phase, Y: yellow phase, O: orange phase, DO: deep orange sicklerite. Plane-polarized light; the widths of the photographs are 800 and 1600 μm , respectively.

were previously described by Wang *et al.* (1981), who investigated the X-ray powder diffraction, optical and thermal properties of the phosphates.

EXPERIMENTAL

An electron-microprobe analysis (Table 1) was done with a Cameca SX-50 instrument (Ruhr-Universität Bochum, Germany) operating in the wavelength-dispersion mode with an accelerating voltage of 15 kV and a beam current of 15 nA. The following standards were used: graffonite (P, Mn, Fe), pyrope (Mg), ZnO (Zn), and andradite (Ca). The counting time was 20 s for all elements.

The Li₂O contents (Table 1) were determined with a Cameca IMS-4f ion microprobe (SIMS) installed at CNR-IGG, Pavia, Italy. We used a 12.5 kV accelerated ¹⁶O⁻ primary-ion beam with a current intensity in the range 0.8–4 nA, corresponding to a beam diameter of ~3–6 mm. The samples were polished, washed in an ultrasonic tank with ethanol, and Pt-coated (400 Å thickness) before analysis. Secondary-ion signals of the isotopes ⁶Li⁺, ³¹P⁺ and ⁵⁷Fe⁺ were detected at the electron multiplier. Acquisition times were 3 s for Li and P (each), and 6 s for Fe over three cycles. Analysis was done under steady-state sputtering conditions after sputtering for 360 s using ~75–125 eV secondary ions. The choice of medium- to high-energy secondary ions (energy filtering) as analytical ones is particularly useful to reduce matrix effects affecting light-element ionization and to improve the reproducibility of analysis (Ottolini *et al.* 1993). In the present case, the precision of analyses, as tested on various (CNR-IGG) Li-bearing homogeneous glassy standards, was found to be better or on the order of 1% rel. (as 1σ) over a one-day analytical span. In order to convert the (⁶Li⁺/³¹P⁺ and ⁶Li⁺/⁵⁷Fe⁺) ion signals into (Li) concentrations, we used as reference material for SIMS analysis of our material for triphylite from the Buranga pegmatite, Rwanda; more details concerning the SIMS analytical procedure are given by Hatert *et al.* (2011a).

The X-ray study of the structures was done on an Oxford Diffraction Gemini PX Ultra four-circle diffractometer equipped with a Ruby CCD area-detector (Namur, Belgium), on crystal fragments drilled out of the thin section (Fig. 1b). The following numbers were used for the crystals: crystal 1: colorless lithiophilite, crystal 2: light yellow phase, crystal 3: yellow phase, crystal 4: orange phase, and crystal 5: deep orange sicklerite. Between 191 (crystal 1) and 1276 (crystal 3) frames with a spatial resolution of 1° were collected by the φ-ω scan technique, with a counting time of 5 to 20 s per frame, in the range 7.82° < 2θ < 85.50°. The number of reflections extracted from these frames, as well as the unit-cell parameters of the phosphates, are given in Table 2. Data were corrected for Lorenz, polariza-

tion and absorption effects, the latter with an empirical method using the SCALE3 ABSPACK scaling algorithm included in the CRYSLISRED package (Oxford Diffraction 2007).

The structures were refined with SHELXTL (Sheldrick 2008) in space group *Pbnm*, starting from the atom coordinates of lithiophilite used by Losey *et al.* (2004), and scattering curves for neutral atoms, together with anomalous dispersion corrections, taken from the *International Tables for X-ray Crystallography, Vol. C* (Wilson 1992). For the sake of simplicity, Mg, Zn and Ca, which occur in low to trace amounts, were not taken into account in crystal-structure refinement. The relative occupancies of Li and vacancies at *M1*, and of Fe and Mn at *M2*, were refined, with the Mn content of the *M2* site constrained to 0.800, according to the electron-microprobe analyses (see below). The refinements were completed using anisotropic-displacement parameters for all atoms. The final conventional *R*₁ factors [*F*₀ > 2σ (*F*₀)] are between 0.0218 and 0.0292; further details of intensity data collection and structure refinement are given in Table 2. A listing of structure factors and a cif file for each refined structure are available from the Depository of Unpublished Data in the Mineralogical Association of Canada website [document Lithiophilite–Sicklerite CM50_843].

TABLE 1. COMPOSITION OF PHOSPHATES OF THE LITHIOPHILITE–SICKLERITE SERIES

Sample <i>n</i>	1 4	2 6	3 4	4 8	5 6
P ₂ O ₅ wt.%	46.93	46.58	46.82	47.06	47.08
Fe ₂ O ₃ *	6.69	3.97	8.44	8.54	8.60
Mn ₂ O ₃ *	-	-	0.34	3.12	9.86
MgO	0.10	0.11	0.07	0.08	0.09
ZnO	0.07	0.07	0.07	0.04	0.05
FeO*	1.72	4.34	-	-	-
MnO*	37.59	37.46	37.72	35.47	29.33
CaO	0.05	0.10	0.03	0.05	0.03
Li ₂ O**	9.21	9.44	8.66	8.11	6.85
Total	102.36	102.07	102.15	102.47	101.89
<i>P apfu</i>	1.000	1.000	1.000	1.000	1.000
Fe ³⁺	0.127	0.076	0.160	0.161	0.162
Mn ³⁺	-	-	0.006	0.060	0.188
Mg	0.004	0.004	0.003	0.003	0.003
Zn	0.001	0.001	0.001	0.001	0.001
Fe ²⁺	0.036	0.092	-	-	-
Mn ²⁺	0.801	0.805	0.806	0.754	0.623
Ca	0.001	0.003	0.001	0.001	0.001
Li	0.932	0.963	0.879	0.819	0.691
Fe/(Fe + Mn)	0.169	0.173	0.165	0.165	0.167

Analyst: H.-J. Bernhardt, of Bochum, Germany, carried out the electron-microprobe analyses. Cation numbers were calculated on the basis of one P atom per formula unit. * The FeO/Fe₂O₃ and MnO/Mn₂O₃ values were calculated to maintain charge balance. ** The Li₂O content was determined by SIMS (analyst: Luisa Ottolini). *n*: number of analyses made.

TABLE 2. EXPERIMENTAL DETAILS FOR THE SINGLE-CRYSTAL X-RAY-DIFFRACTION STUDY OF THE LITHIOPHILITE-SICKLERITE SERIES

Sample	1	2	3	4	5
Color	Colorless	Light yellow	Yellow	Orange	Deep orange
Dimensions of the crystal (mm)	ca. 0.08 × 0.08 × 0.03	ca. 0.08 × 0.08 × 0.03	ca. 0.08 × 0.08 × 0.03	ca. 0.08 × 0.08 × 0.03	ca. 0.08 × 0.08 × 0.03
a (Å)	4.736(1)	4.734(1)	4.740(1)	4.767(1)	4.765(1)
b (Å)	10.432(2)	10.423(2)	10.415(1)	10.403(2)	10.338(1)
c (Å)	6.088(1)	6.094(1)	6.080(1)	6.072(1)	6.060(1)
Space group	<i>Pbnm</i>	<i>Pbnm</i>	<i>Pbnm</i>	<i>Pbnm</i>	<i>Pbnm</i>
Z	4	4	4	4	4
2θ _{min} , 2θ _{max}	7.82°, 65.20°	7.82°, 65.16°	7.82°, 85.50°	7.84°, 64.88°	7.88°, 85.50°
Range of indices	-7 ≤ h ≤ 3, -8 ≤ k ≤ 15, -9 ≤ l ≤ 7	-7 ≤ h ≤ 6, -7 ≤ k ≤ 15, -8 ≤ l ≤ 7	-8 ≤ h ≤ 7, -18 ≤ k ≤ 18, -10 ≤ l ≤ 10	-6 ≤ h ≤ 7, -8 ≤ k ≤ 15, -6 ≤ l ≤ 9	-8 ≤ h ≤ 7, -19 ≤ k ≤ 18, -10 ≤ l ≤ 10
Measured intensities	1539	1653	5279	1515	5167
Unique reflections	545	548	1095	547	1066
Independent non-zero [<i>I</i> > 2σ(<i>I</i>)] reflections	480	492	891	464	876
μ (mm ⁻¹)	4.862	4.895	4.872	4.888	4.899
Refined parameters	42	42	42	42	42
R ₁ [<i>F</i> _o > 2σ(<i>F</i> _c)]	0.0235	0.0219	0.0292	0.0250	0.0218
R ₁ (all)	0.0278	0.0260	0.0376	0.0320	0.0298
wR ₂ (all)	0.0622	0.0580	0.0635	0.0605	0.0541
S (goodness of fit)	1.003	0.949	1.062	0.992	1.052
Max. Δσ in the last least-squares cycle	0.000	0.000	0.000	0.000	0.000
Max. peak and hole in the final ΔF map (e/Å ³)	+0.64 and -0.60	+0.66 and -0.51	+1.89 and -0.81	+0.83 and -0.51	+1.19 and -0.56

CHEMICAL VARIATIONS ALONG THE LITHIOPHILITE-SICKLERITE SERIES

Under the polarizing microscope, lithiophilite is colorless, whereas sicklerite is deep orange. Several grains also show intermediate colors (Fig. 1), suggesting a progressive transition from lithiophilite to sicklerite. The SIMS results for Li₂O (wt.%) are: Sample 1 (four analyses): 9.21 ± 0.11 (1.2 as 1σ); sample 2 (four analyses): 9.44 ± 0.04 (0.5 as 1σ); sample 3 (six analyses): 8.66 ± 0.56 (6.5 as 1σ); sample 4 (six analyses): 8.11 ± 0.30 (3.7 as 1σ); sample 5 (six analyses): 6.85 ± 0.35 (5.0 as 1σ). These data indicate a higher micro-scale inhomogeneity of Li in samples 3, 4, and 5. Nevertheless, the progressive transition observed optically is confirmed by results of the SIMS analyses, which indicate Li values from 0.963 (lithiophilite) to 0.691 (sicklerite) Li atoms per formula unit (*apfu*) (Table 1). The charge deficit induced by this leaching of Li is compensated by progressive oxidation of Fe²⁺ to Fe³⁺ (samples 1 to 3), followed by oxidation of Mn²⁺ to Mn³⁺ (samples 3 to 5; Table 1). Moreover, it seems that the presence of Mn³⁺ in sicklerite is correlated with the strong orange color of the samples, clearly visible in crystals 3, 4, and 5 (Fig. 1).

It is noteworthy that this progressive oxidation does not induce significant variations of Fe/(Fe + Mn), which ranges between 0.165 and 0.173 (Table 1). This constancy of the Fe/(Fe + Mn) value is well established

during the oxidation of triphylite (*e.g.*, Fransolet *et al.* 1985, 1986, Roda *et al.* 1996, 2004, Roda-Robles *et al.* 1998, 2010), and is due to the topotactic nature of this oxidation process, which preserves the olivine structure-type of the parent phosphates.

STRUCTURE REFINEMENT

Final positional and equivalent isotropic-displacement parameters are given in Table 3, and selected interatomic distances and angles are given in Table 4. The structure is identical to that of triphylite (Losey *et al.* 2004), and is characterized by two chains of edge-sharing octahedra parallel to the *c* axis: a first chain consists of *M1* octahedra containing Li atoms and vacancies, and the second chain is formed by the *M2* sites, occupied by Fe and Mn. The chains are connected in the *b* direction by a sharing of edges of the octahedra, and the resulting sheets are connected in the *a* direction by PO₄ tetrahedra (Fig. 2).

A detailed distribution of cations has also been established by taking into account the results of the chemical analyses and of the single-crystal structure refinement. The results given in Table 5 indicate that the refined site-populations (RSP) obtained from the single-crystal structure refinement are in good agreement with the assigned site-populations (ASP) deduced from the electron-microprobe and SIMS results. Moreover, the refined site-scattering values (RSS) and the mean bond-

lengths (MBL) obtained from the structure refinement are very close to the calculated site-scattering values (CSS) and the calculated bond-lengths (CBL), respectively (Table 5). This agreement again confirms the reliability of the site populations assigned.

Finally, the bond-valence sums were calculated as $s = \exp[(R_0 - R)/0.37]$, by using the R_0 values of Brown & Altermatt (1985). The sums for O and P atoms are in the ranges 1.84–2.02 and 4.90–4.93 *vu*, respectively, in good agreement with their ideal values. For the $M2$ site, the bond-valence sums show a significant increase from 2.01 (crystal 1) to 2.12 *vu* (crystal 5) (Fig. 3b), due to the progressive oxidation of Fe and Mn at that site, which produces a decrease of the mean $M2$ –O bond

lengths (Table 4). Conversely, a decrease in the sums is observed at the $M1$ site, from 0.93 (crystal 1) to 0.87 *vu* (crystal 5) (Fig. 3a), correlated with an increase of the $M1$ –O bond lengths (Table 4). This decrease of the bond-valence sum at the $M1$ site is due to decrease of the Li occupancy at that site (Table 3).

DISCUSSION

Crystal chemistry of the lithiophilite–sicklerite series

Variations in the unit-cell parameters along the triphylite–lithiophilite series were investigated by X-ray powder diffraction (Fransolet *et al.* 1984) and single-crystal X-ray diffraction (Losey *et al.* 2004); these studies show a significant increase of the unit-cell parameters as Fe/(Fe + Mn) decreases. In the lithiophilite–sicklerite series, a different pattern of behavior is observed, with a decrease in b and c , but an increase in a , as the lithium content decreases (Fig. 4). The increase in a can be explained by the increase of the $M1$ –O1 and $M1$ –O2 bond lengths, which form a square parallel to the a axis (Table 4, Fig. 5a), whereas the decrease of b is correlated with a decrease of the $M2$ –O1 and $M2$ –O2 bond lengths, which are parallel to the b axis (Table 4, Fig. 5b).

In the triphylite–lithiophilite series, Losey *et al.* (2004) observed a decrease of the $M2$ –O bond lengths as Mn^{2+} (effective ionic radius, *e.i.r.*, 0.830 Å; Shannon 1976) is replaced by Fe^{2+} (*e.i.r.* 0.780 Å; Shannon 1976). This decrease is positively correlated with a decrease of the $M1$ –O bond lengths, except for the $M1$ –O3 bonds, which increase significantly. As Fe and Mn occur at the $M2$ site in olivine-type phosphates, the decrease of $M2$ –O bond lengths was expected by Losey *et al.* (2004), but the decrease of $M1$ –O bond lengths is more difficult to understand because this site is solely occupied by Li in this series. In the lithiophilite–sicklerite series, a global decrease of the $M2$ –O bond lengths is observed, produced by oxidation of Fe^{2+} and Mn^{2+} to Fe^{3+} (*e.i.r.* 0.645 Å; Shannon 1976) and Mn^{3+} (*e.i.r.* 0.645 Å; Shannon 1976), respectively (Fig. 6b). Simultaneous with this oxidation, the lithium content decreases at $M1$, inducing an increase of the $M1$ –O distances necessary to satisfy bond-valence requirements (Fig. 6a). As a consequence, the $M1$ –O and $M2$ –O bond distances negatively correlate with each other in the lithiophilite–sicklerite series, whereas they are positively correlated to each other in the triphylite–lithiophilite series. The negative correlation observed in the lithiophilite–sicklerite series is governed mainly by charge-balance requirements, which necessitate a decrease in the Li content at $M1$ in order to compensate the oxidation of Fe and Mn at $M2$. This correlation is also certainly influenced by crystal-chemical constraints: indeed, the olivine structure is extremely compact, and each oxygen atom is bonded to both $M1$ and $M2$ cations, thus explaining why a volume increase

TABLE 3. FINAL FRACTIONAL COORDINATES AND EQUIVALENT DISPLACEMENT PARAMETERS (\AA^2) OF ATOMS IN PHOSPHATES OF THE LITHIOPHILITE–SICKLERITE SERIES

Site	Atom	x	y	z	U_{eq}
Sample 1: Colorless					
M1	Li*	0	0	0	0.016(2)
M2	Mn, Fe*	0.97196(9)	0.28174(4)	¼	0.0072(1)
P	P	0.4104(1)	0.09276(7)	¼	0.0065(2)
O1	O	0.7321(4)	0.0969(2)	¼	0.0099(4)
O2	O	0.2111(4)	0.4560(2)	¼	0.0088(4)
O3	O	0.2776(3)	0.1621(1)	0.0490(2)	0.0095(3)
Sample 2: Light yellow					
M1	Li [†]	0	0	0	0.014(2)
M2	Mn, Fe*	0.97198(8)	0.28175(3)	¼	0.0081(1)
P	P	0.4103(1)	0.09283(6)	¼	0.0076(2)
O1	O	0.7318(4)	0.0971(2)	¼	0.0114(4)
O2	O	0.2106(3)	0.4559(2)	¼	0.0101(3)
O3	O	0.2784(2)	0.1621(1)	0.0490(2)	0.0103(3)
Sample 3: Yellow					
M1	Li*	0	0	0	0.017(1)
M2	Mn, Fe*	0.97167(6)	0.28160(2)	¼	0.00845(7)
P	P	0.41062(8)	0.09310(4)	¼	0.0071(1)
O1	O	0.7308(2)	0.0981(1)	¼	0.0115(2)
O2	O	0.2081(2)	0.4557(1)	¼	0.0103(2)
O3	O	0.2772(2)	0.16258(7)	0.0490(1)	0.0109(2)
Sample 4: Orange					
M1	Li*	0	0	0	0.019(2)
M2	Mn, Fe*	0.97049(9)	0.28133(4)	¼	0.0079(1)
P	P	0.4108(1)	0.09375(7)	¼	0.0079(2)
O1	O	0.7299(5)	0.1007(2)	¼	0.0137(4)
O2	O	0.2030(4)	0.4555(2)	¼	0.0126(4)
O3	O	0.2763(3)	0.1637(2)	0.0497(2)	0.0131(3)
Sample 5: Deep orange					
M1	Li*	0	0	0	0.022(2)
M2	Mn, Fe*	0.97002(5)	0.28118(2)	¼	0.00931(6)
P	P	0.41101(8)	0.09406(3)	¼	0.00849(9)
O1	O	0.7292(3)	0.1021(1)	¼	0.0150(2)
O2	O	0.2001(3)	0.45515(9)	¼	0.0136(2)
O3	O	0.2759(2)	0.16404(7)	0.0492(1)	0.0142(2)

* The occupancy factors for the $M1$ and $M2$ sites are as follows. Sample 1: 0.91(3) Li and 0.800 Mn + 0.183(3) Fe; sample 2: 0.86(3) Li and 0.800 Mn + 0.183(3) Fe; sample 3: 0.84(2) Li and 0.800 Mn + 0.200(2) Fe; sample 4: 0.76(3) Li and 0.800 Mn + 0.180(3) Fe; sample 5: 0.75(2) Li and 0.800 Mn + 0.195(2) Fe.

TABLE 4. SELECTED BOND-DISTANCES (Å) AND ANGLES (°) FOR PHOSPHATES OF THE LITHIOPHILITE-SICKLERITE SERIES

Sample	1	2	3	4	5	Difference
<i>M1-O1</i> × 2	2.224(1)	2.227(1)	2.2319(9)	2.249(2)	2.2527(9)	+0.029
<i>M1-O2</i> × 2	2.098(1)	2.100(1)	2.1065(8)	2.127(2)	2.1334(9)	+0.035
<i>M1-O3</i> × 2	2.163(1)	2.164(1)	2.1639(9)	2.174(2)	2.1663(8)	+0.003
Mean	2.162	2.164	2.167	2.183	2.184	+0.022
<i>M2-O1</i>	2.238(2)	2.235(2)	2.226(1)	2.201(2)	2.178(1)	-0.060
<i>M2-O2</i>	2.142(2)	2.138(2)	2.132(1)	2.124(2)	2.106(1)	-0.036
<i>M2-O3</i> × 2	2.122(1)	2.122(1)	2.1194(8)	2.120(1)	2.1127(8)	-0.009
<i>M2-O3'</i> × 2	2.270(1)	2.271(1)	2.2644(9)	2.259(2)	2.2519(8)	-0.018
Mean	2.194	2.193	2.188	2.181	2.169	-0.025
P-O1	1.524(2)	1.523(2)	1.519(1)	1.523(2)	1.518(1)	-0.006
P-O2	1.538(2)	1.538(2)	1.537(1)	1.537(2)	1.531(1)	-0.007
P-O3 × 2	1.554(1)	1.553(1)	1.5546(8)	1.555(2)	1.5551(7)	+0.001
Mean	1.543	1.542	1.541	1.543	1.540	-0.003
O1-P-O2	113.5(1)	113.5(1)	113.43(6)	113.4(1)	113.39(6)	
O1-P-O3 × 2	113.04(7)	112.85(6)	112.99(4)	112.92(7)	112.82(4)	
O2-P-O3 × 2	106.28(7)	106.37(6)	106.52(4)	106.99(8)	107.05(4)	
O3-P-O3	103.9(1)	104.1(1)	103.63(7)	102.9(1)	102.96(6)	
Mean	109.34	109.34	109.35	109.35	109.35	

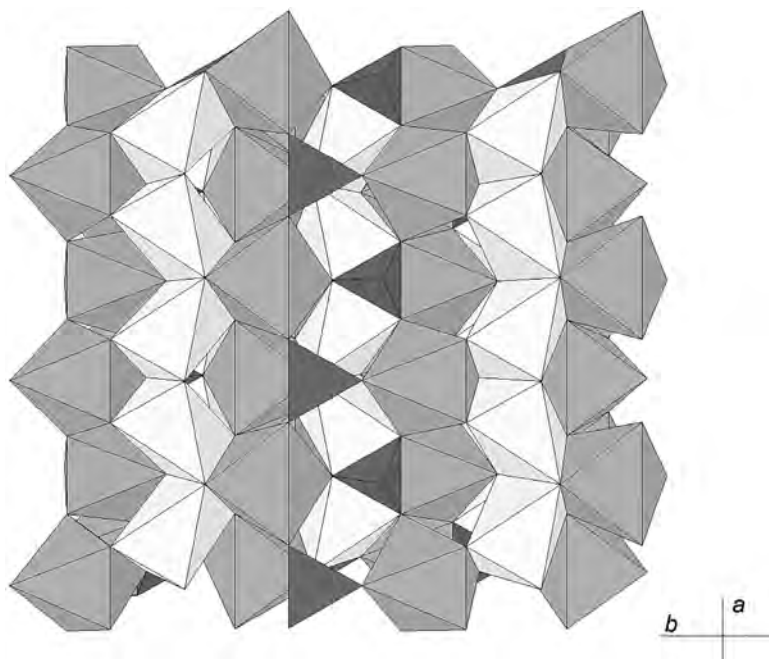


FIG. 2. The crystal structure of sicklerite, projected perpendicular to the *c* axis. The *M1* octahedra containing Li are white, the *M2* octahedra containing Fe and Mn are light grey, and the PO_4 tetrahedra are dark grey.

of one of these sites implies a volume decrease of the adjacent polyhedron.

Finally, bond-length distortion (BLD) for the $M1$ and $M2$ sites was calculated with the formula of Renner & Lehmann (1986). As shown on Figure 7, the BLD coefficient for $M1$ decreases linearly with increasing Li content (Fig. 7a), whereas the BLD coefficient on $M2$ shows the inverse behavior (Fig. 7b). The low BLD coefficient for the $M2$ site of sicklerite is unexpected, as Eventoff *et al.* (1972) noticed that the complete oxidation of iron and manganese induces a Jahn–Teller distortion of the $M(2)O_6$ octahedron in heterosite, characterized by two short bonds (1.912–1.914 Å) and four long bonds (2.030–2.163 Å). Such a strong distortion of the $M2$ octahedron is not observed in sicklerite, but the presence of larger amounts of Mn^{3+} at that site in

purpurite would certainly increase significantly the BLD, as shown by Hatert *et al.* (2011b) in the series triphylite – ferrisicklerite – heterosite. Conversely, the increase of the BLD coefficient of $M1$, observed as the Li content decreases (Fig. 7a), is certainly due to the partial occupancy of that site. Indeed, the average ionic radius at $M1$ decreases significantly as the Li content decreases, owing to the presence of vacancies, and the coordination polyhedron consequently becomes too large. According to the distortion theorem (Brown 2002), any anion located in too large a cavity will induce a distortion of its environment in order to increase the bond-valence sum to the expected value, thus explaining the increase of $M1$ BDL in Li-poor members of the lithiophilite–sicklerite series (Fig. 7a).

Chemistry, nomenclature, and stability of sicklerite

The lithium contents determined by SIMS in members of the lithiophilite–sicklerite series (Table 1) are in good agreement with the Li values determined by refinement of the $M1$ site-occupancy factors (Table 3), thus confirming the reliability of the SIMS analytical procedure applied here. A comparison of the Li content of sicklerite from the Altai Mountains with those of sicklerite from Pala, San Diego County, California (Schaller 1912), Wodgina, Western Australia, and Eräjärvi, Finland (Mason 1941), shows a higher Li content for the sample investigated here (Table 1). As a consequence, we take note of the unexpected variability of the Li content in sicklerite, which ranges from 6.85 to 3.80 wt.% Li_2O . Moreover, the existence of intermediate phases, characterized by Li contents between 9.44 and 8.11 wt.% Li_2O , raises an important nomenclature question: where is the boundary between sicklerite and lithiophilite? According to Fontan *et al.* (1976), the distinction between these species is based mainly on their optical properties and their color; consequently, crystals 1 and 2, showing light colors (Fig. 1), can be considered as lithiophilite, whereas crystals 3, 4, and 5, with strong colors and containing significant amounts of Mn^{3+} (Fig. 1, Table 1), correspond to sicklerite. As these minerals were historically defined on the basis of their optical properties, it is difficult to apply the dominant-constituent rule in this case (Hatert & Burke 2008). However, new analyses on these phosphates are being conducted in order to elucidate this delicate question in nomenclature issues.

The interpretation of the electron-microprobe analyses of sicklerite is based on the assumption that the mineral contains Fe^{3+} and Mn^{2+} , but no Mn^{3+} . This hypothesis, from which Li contents of sicklerite samples are estimated, is clearly ruled out by the chemical data presented here, which show up to 9.86 wt.% Mn_2O_3 in crystal 5 (Table 1). Moreover, the presence of significant amounts of Mn^{3+} in sicklerite was already mentioned by Fontan *et al.* (1976), and the wet-chemical analyses reported by Palache *et al.* (1951) show Mn^{3+} contents

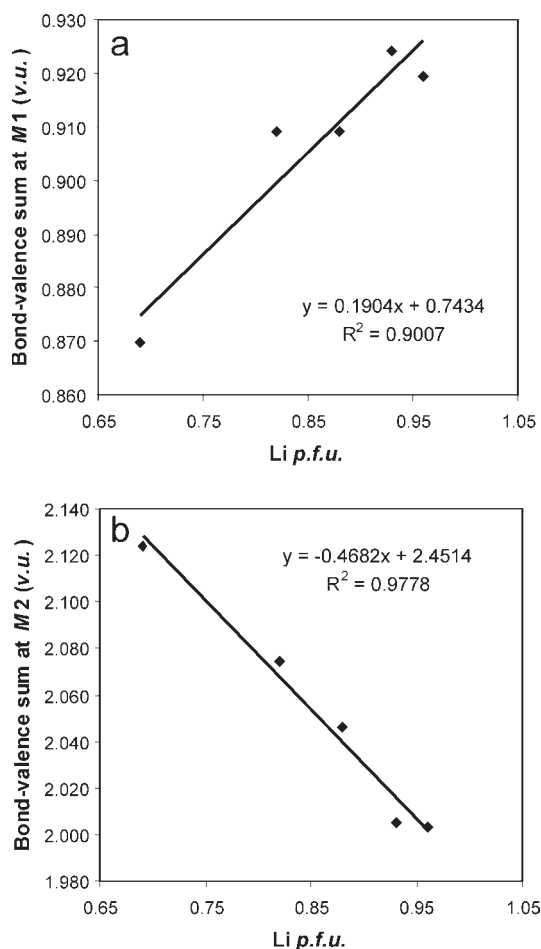


FIG. 3. Variations of bond-valence sums at the $M1$ (a) and $M2$ (b) crystallographic sites, along the lithiophilite–sicklerite series.

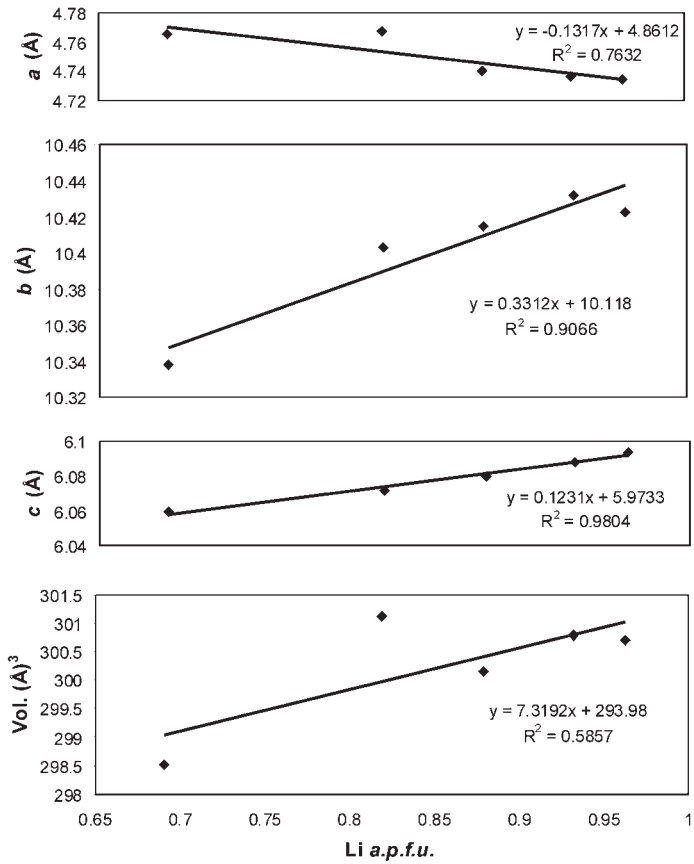


FIG. 4. Variations of the unit-cell parameters along the lithiophilite-sicklerite series.

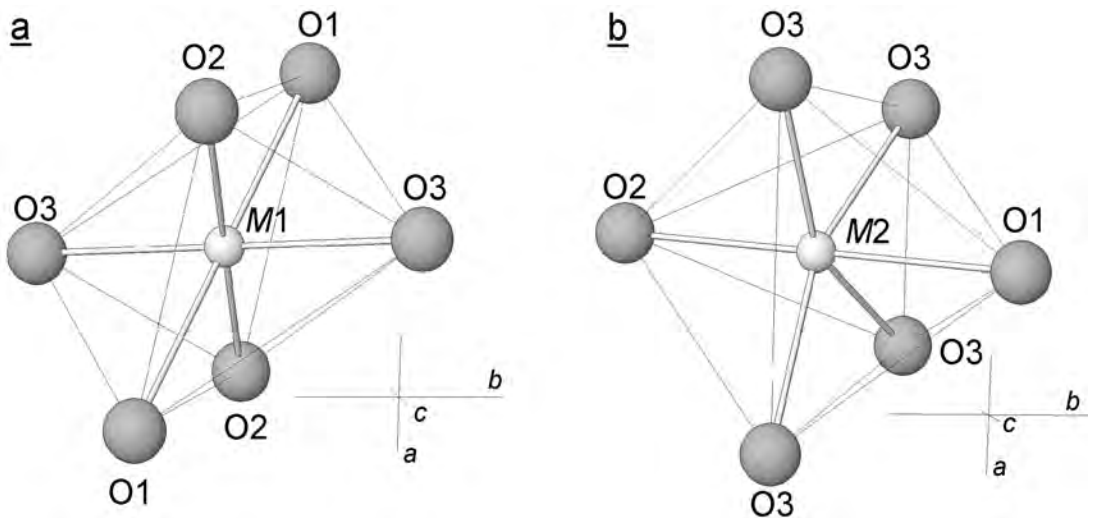


FIG. 5. Morphologies of the M1 (a) and M2 (b) octahedrally coordinated sites in the crystal structure of sicklerite.

TABLE 5. REFINED SITE-POPULATIONS (RSP, *apfu*), REFINED SITE-SCATTERING VALUES (RSS, *epfu*), MEAN BOND-LENGTHS (MBL, Å), ASSIGNED SITE-POPULATIONS (ASP, *apfu*), CALCULATED SITE-SCATTERING VALUES (CSS, *epfu*), AND CALCULATED BOND-LENGTHS (CBL, Å) IN PHOSPHATES OF THE LITHIOPHILITE–SICKLERITE SERIES

Site	Results of the structure determination			Results of the chemical analysis		
	RSP	RSS	MBL	ASP	CSS	CBL*
Sample 1: Colorless						
M1	0.91(3) Li	2.7	2.162	0.932 Li	2.8	2.160
M2	0.800 Mn + 0.183(3) Fe	24.8	2.194	0.800 Mn ²⁺ + 0.150 Fe ³⁺ + 0.050 Fe ²⁺	25.2	2.200
Sample 2: Light yellow						
M1	0.86(3) Li	2.6	2.164	0.963 Li	2.9	2.160
M2	0.800 Mn + 0.183(3) Fe	24.8	2.193	0.800 Mn ²⁺ + 0.100 Fe ³⁺ + 0.100 Fe ²⁺	25.2	2.207
Sample 3: Yellow						
M1	0.84(2) Li	2.5	2.168	0.879 Li	2.6	2.160
M2	0.800 Mn + 0.200(2) Fe	25.2	2.188	0.800 Mn ²⁺ + 0.150 Fe ³⁺ + 0.050 Mn ³⁺	25.2	2.193
Sample 4: Orange						
M1	0.76(3) Li	2.3	2.183	0.819 Li	2.5	2.160
M2	0.800 Mn + 0.180(3) Fe	24.7	2.181	0.750 Mn ²⁺ + 0.150 Fe ³⁺ + 0.100 Mn ³⁺	25.2	2.184
Sample 5: Deep orange						
M1	0.75(2) Li	2.2	2.184	0.691 Li	2.1	2.160
M2	0.800 Mn + 0.195(2) Fe	25.1	2.169	0.650 Mn ²⁺ + 0.150 Fe ³⁺ + 0.200 Mn ³⁺	25.2	2.165

* The CBL values have been calculated from the ASP with the effective ionic radii of Shannon (1976), assuming a full occupancy of the crystallographic sites.

between 2.10 and 8.19 wt.% Mn₂O₃. Consequently, the Li contents of sicklerite samples estimated from electron-microprobe data are not reliable.

As underlined by Fontan *et al.* (1976), the presence of significant amounts of Mn³⁺ in sicklerite, and its near-absence from ferrisicklerite, indicate different processes of formation for these two phosphates. This distinction is confirmed by the petrographic and chemical observations given in the present paper, which clearly show a progressive transition from lithiophilite to sicklerite (Fig. 1, Table 1), never observed in the triphylite–ferrisicklerite series. These different processes of oxidation, certainly controlled by the distinct geochemical behaviors of iron and manganese, are also responsible for the thermal instability of synthetic purpurite, which transforms to Mn₂P₂O₇ above 210°C, whereas synthetic heterosite, its Fe³⁺-dominant analogue, is stable up to a high temperature (Kim *et al.* 2009, Chen & Richardson 2010).

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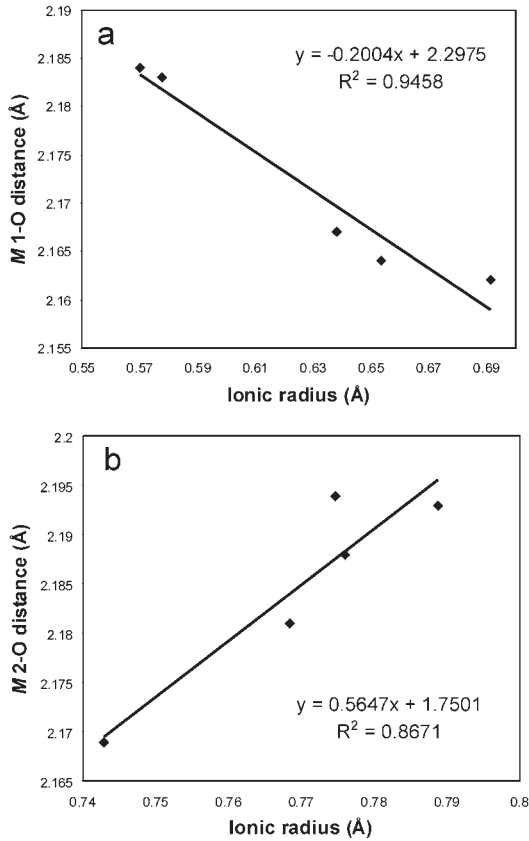


FIG. 6. Correlations between the $M-O$ distances and the mean ionic radii of the cations occurring at the $M1$ (a) and $M2$ (b) sites in the lithiophilite-sicklerite series.

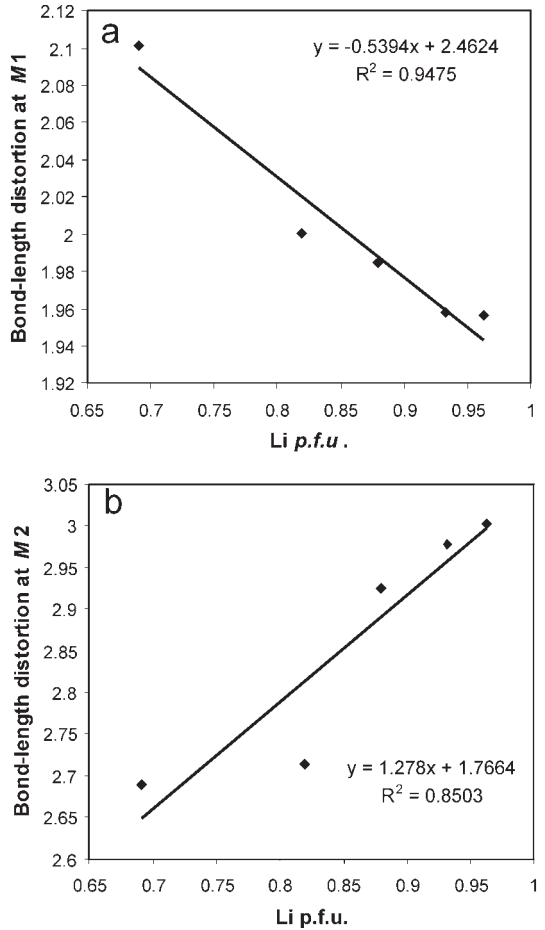


FIG. 7. Variations of bond-length distortion at $M1$ (a) and $M2$ (b) along the lithiophilite-sicklerite series.

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_refine_special_details
;
Refinement of F2 against ALL reflections. The weighted R-factor wR and
goodness of fit S are based on F2, conventional R-factors R are based
on F, with F set to zero for negative F2. The threshold expression of
F2 > 2sigma(F2) is used only for calculating R-factors(gt) etc. and is
not relevant to the choice of reflections for refinement. R-factors based
on F2 are statistically about twice as large as those based on F, and R-
factors based on ALL data will be even larger.
;

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P 0.0059(3) 0.0060(3) 0.0075(3) 0.000 0.000 -0.0002(2)
O1 0.0056(8) 0.0103(9) 0.0137(9) 0.000 0.000 -0.0009(7)
O2 0.0082(8) 0.0070(8) 0.0112(9) 0.000 0.000 0.0010(7)
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_geom_special_details

;
All esds (except the esd in the dihedral angle between two l.s. planes)
are estimated using the full covariance matrix. The cell esds are taken
into account individually in the estimation of esds in distances, angles
and torsion angles; correlations between esds in cell parameters are only
used when they are defined by crystal symmetry. An approximate (isotropic)
treatment of cell esds is used for estimating esds involving l.s. planes.

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Li1 O1 2.2242(14) 5_655 ?
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Li1 P 2.6514(6) 5 ?
Li1 P 2.6514(6) . ?
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Li1 Li1 3.0440(6) 3 ?
Li1 Mn2 3.3125(6) 1_455 ?
Li1 Mn2 3.3125(6) 5_655 ?
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Mn2 O3 2.2695(14) 1_655 ?
Mn2 O3 2.2695(14) 7_656 ?
Mn2 P 2.8631(9) 1_655 ?
Mn2 Li1 3.3125(6) 1_655 ?

Mn2 Li1 3.3125(6) 3_655 ?
P O1 1.524(2) . ?
P O2 1.5382(19) 6_655 ?
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O2 Li1 O1 88.57(5) 6_655 1_455 ?
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O3 Li1 O1 84.10(6) . 1_455 ?
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O1 Li1 P 98.07(5) 1_455 5 ?
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P Li1 Mn2 123.932(17) . 5_655 ?
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'-x, -y, -z'
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_refine_special_details
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Refinement of F2 against ALL reflections. The weighted R-factor wR and
goodness of fit S are based on F2, conventional R-factors R are based
on F, with F set to zero for negative F2. The threshold expression of
F2 > 2sigma(F2) is used only for calculating R-factors(gt) etc. and is
not relevant to the choice of reflections for refinement. R-factors based
on F2 are statistically about twice as large as those based on F, and R-
factors based on ALL data will be even larger.
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_refine_ls_weighting_details
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P P 0.41030(12) 0.09283(6) 0.2500 0.00756(17) Uani 1 2 d S . .

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Mn2 0.00910(19) 0.00719(19) 0.0080(2) 0.000 0.000 0.00057(12)
P 0.0071(3) 0.0074(3) 0.0081(3) 0.000 0.000 0.0002(2)
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O2 0.0090(7) 0.0079(7) 0.0134(8) 0.000 0.000 0.0015(6)
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_geom_special_details

;
All esds (except the esd in the dihedral angle between two l.s. planes)
are estimated using the full covariance matrix. The cell esds are taken
into account individually in the estimation of esds in distances, angles
and torsion angles; correlations between esds in cell parameters are only
used when they are defined by crystal symmetry. An approximate (isotropic)
treatment of cell esds is used for estimating esds involving l.s. planes.

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Li1 P 2.6514(6) 5 ?
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Li1 Li1 3.0470(6) 3_554 ?
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Li1 Mn2 3.3110(6) 5_655 ?
Li1 Mn2 3.3110(6) 1_455 ?
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Mn2 O2 2.1382(18) 1_655 ?
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Mn2 O3 2.2712(13) 7_656 ?
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P Mn2 2.8606(8) 1_455 ?
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O3 P O3 104.12(10) . 7_556 ?
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O1 P Mn2 134.82(8) . 1_455 ?
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'-x, -y, -z'
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Refinement of F^2 against ALL reflections. The weighted R-factor wR and goodness of fit S are based on F^2 , conventional R-factors R are based on F , with F set to zero for negative F^2 . The threshold expression of $F^2 > 2\sigma(F^2)$ is used only for calculating R-factors(gt) etc. and is not relevant to the choice of reflections for refinement. R-factors based on F^2 are statistically about twice as large as those based on F , and R-factors based on ALL data will be even larger.

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P 0.00666(17) 0.00712(16) 0.00762(17) 0.000 0.000 0.00019(11)
O1 0.0078(5) 0.0140(5) 0.0127(5) 0.000 0.000 -0.0004(4)
O2 0.0100(5) 0.0080(4) 0.0130(5) 0.000 0.000 0.0009(4)
O3 0.0101(3) 0.0130(3) 0.0098(3) 0.0033(3) 0.0003(3) 0.0019(3)

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;
All esds (except the esd in the dihedral angle between two l.s. planes)
are estimated using the full covariance matrix. The cell esds are taken
into account individually in the estimation of esds in distances, angles
and torsion angles; correlations between esds in cell parameters are only
used when they are defined by crystal symmetry. An approximate (isotropic)
treatment of cell esds is used for estimating esds involving l.s. planes.

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Li1 O2 2.1065(8) 2_455 ?
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Li1 O3 2.1639(9) 5 ?
Li1 O1 2.2319(9) 5_655 ?
Li1 O1 2.2319(9) 1_455 ?
Li1 P 2.6531(4) . ?
Li1 P 2.6531(4) 5 ?
Li1 Li1 3.0398(3) 3 ?
Li1 Li1 3.0398(3) 3_554 ?
Li1 Mn2 3.3059(3) 1_455 ?
Li1 Mn2 3.3059(3) 5_655 ?
Mn2 O3 2.1194(8) 2 ?
Mn2 O3 2.1194(8) 8_666 ?
Mn2 O2 2.1318(11) 1_655 ?
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Mn2 O3 2.2644(9) 1_655 ?

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O2 Li1 P 35.38(3) 2_455 5 ?
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O3 Li1 P 35.87(2) 5 5 ?
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P Li1 P 180.000(15) . 5 ?
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Refinement of F2 against ALL reflections. The weighted R-factor wR and
goodness of fit S are based on F2, conventional R-factors R are based
on F, with F set to zero for negative F2. The threshold expression of
F2 > 2sigma(F2) is used only for calculating R-factors(gt) etc. and is
not relevant to the choice of reflections for refinement. R-factors based
on F2 are statistically about twice as large as those based on F, and R-
factors based on ALL data will be even larger.
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P 0.0063(3) 0.0085(3) 0.0089(4) 0.000 0.000 0.0000(2)
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O2 0.0126(10) 0.0099(9) 0.0153(10) 0.000 0.000 0.0002(8)
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All esds (except the esd in the dihedral angle between two l.s. planes)
are estimated using the full covariance matrix. The cell esds are taken
into account individually in the estimation of esds in distances, angles
and torsion angles; correlations between esds in cell parameters are only
used when they are defined by crystal symmetry. An approximate (isotropic)
treatment of cell esds is used for estimating esds involving l.s. planes.

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P Mn2 2.8658(9) 1_455 ?
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Refinement of F2 against ALL reflections. The weighted R-factor wR and
goodness of fit S are based on F2, conventional R-factors R are based
on F, with F set to zero for negative F2. The threshold expression of
F2 > 2sigma(F2) is used only for calculating R-factors(gt) etc. and is
not relevant to the choice of reflections for refinement. R-factors based
on F2 are statistically about twice as large as those based on F, and R-
factors based on ALL data will be even larger.
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_atom_site_occupancy
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Mn2 0.00996(11) 0.00909(9) 0.00887(9) 0.000 0.000 -0.00036(7)
P 0.00722(17) 0.00933(15) 0.00893(15) 0.000 0.000 0.00000(11)
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O2 0.0152(5) 0.0109(4) 0.0147(4) 0.000 0.000 0.0004(4)
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All esds (except the esd in the dihedral angle between two l.s. planes)
are estimated using the full covariance matrix. The cell esds are taken
into account individually in the estimation of esds in distances, angles
and torsion angles; correlations between esds in cell parameters are only
used when they are defined by crystal symmetry. An approximate (isotropic)
treatment of cell esds is used for estimating esds involving l.s. planes.

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